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**Abstract.** Hafnium oxide (HfO<sub>2</sub>) is the most frequently used high-index material in multilayer thin-film coatings for high-power laser applications ranging from near-infrared to near-ultraviolet (UV). Absorption in this high-index material is also known to be responsible for nanosecond-pulse laser-damage initiation in multilayers. In this work, modification of the near-UV absorption of HfO<sub>2</sub> monolayer films subjected to irradiation by continuous-wave (cw), 355-nm or 351-nm laser light focused to produce power densities of the order of  $\sim 100$  kW/cm<sup>2</sup> is studied. Up to a 70% reduction in absorption is found in the areas subjected to irradiation. Temporal behavior of absorption is characterized by a rapid initial drop on the few-tens-of-seconds time scale, followed by a longer-term decline to a steady-state level. Absorption maps generated by photothermal heterodyne imaging confirm the permanent character of the observed effect. Nanosecond-pulse, 351-nm and 600-fs, 1053-nm laser-damage tests performed on these cw laser-irradiated areas confirm a reduction of absorption by measuring up to 25% higher damage thresholds. We discuss possible mechanisms responsible for near-UV absorption annealing and damage-threshold improvement resulting from irradiation by near-UV cw laser light. © The Authors. Published by SPIE under a Creative Commons Attribution 3.0 Unported License. Distribution or reproduction of this work in whole or in part requires full attribution of the original publication, including its DOI. [DOI: [10.1117/1.OE.53.12.122504](https://doi.org/10.1117/1.OE.53.12.122504)]

Keywords: absorption annealing; hafnium oxide thin films; laser damage.

Paper 140509SSP received Mar. 28, 2014; revised manuscript received May 7, 2014; accepted for publication May 28, 2014; published online Jun. 25, 2014.

## 1 Introduction

Hafnium oxide (HfO<sub>2</sub>) is used in numerous applications as a high-index component in coatings for high-power lasers ranging from a near-infrared to near-ultraviolet (UV) light spectrum. It is also known that for the most frequently used film-pair combination of HfO<sub>2</sub>/SiO<sub>2</sub>, HfO<sub>2</sub> is the material where nanosecond-pulse laser damage is initiated.<sup>1</sup> In this context, reduction of absorption in this material is crucial to improving damage performance of the coatings used in mirrors and other laser components. Previous studies<sup>2</sup> showed that near-UV absorption in HfO<sub>2</sub> thin films is generated by high-spatial-density (an average separation of 100 nm or less) nanoscale absorbers whose nature is attributed to the agglomeration of electronic defects. Electronic defects exist even in high-quality optical bulk materials, such as crystals and glasses. In thin films, additional absorbing states might exist because of the presence of interfaces and grain boundaries. There are very limited ways to influence the concentration of the absorbing states once the thin film is deposited. Thermal annealing<sup>3,4</sup> is the process most frequently used as a research tool, but there are obvious practical limitations for optical parts used in large-scale lasers. Irradiation by pulsed laser radiation at fluences below damage threshold (also called laser conditioning) is another widely used method.

In this work, we explore the possibility of using continuous-wave (cw) laser radiation with power densities in the range of 50 kW/cm<sup>2</sup> to  $>1$  MW/cm<sup>2</sup> to anneal absorption in HfO<sub>2</sub> monolayer films in the near-UV spectral range. We also investigate the absorption-annealing impact on

pulsed-laser damage behavior of HfO<sub>2</sub> monolayers subjected to irradiation by 351-nm, 0.9-ns pulses and 1053-nm, 600-fs pulses.

## 2 Experimental

An HfO<sub>2</sub> monolayer film with a 180-nm physical thickness (one wave at 351 nm) was *e*-beam deposited on a fused-silica substrate on top of a 500-nm-thick silicon dioxide (SiO<sub>2</sub>) film, isolating the HfO<sub>2</sub> film from the substrate defects introduced by the polishing process. The deposition rate was 1.2 Å/s and the oxygen pressure was  $8 \times 10^{-5}$  Torr. For monolayer absorption characterization, we used a photothermal heterodyne imaging (PHI) technique, utilizing pump and probe laser beams focused into a submicrometer spot on a sample having the same objective. Modulated pump-light absorption inside the sample produces a locally modulated refractive-index variation, which causes probe-light scattering amplified by far-field interference. Using the nanopositioning stage for the sample translation allows one to map absorption of the film sample with a spatial resolution better than 0.4 μm. Throughout this paper, the absorption will be quantified in terms of the PHI signal (in microvolts), which is proportional to the absorption cross-section. Calibration of absolute absorption values using photothermal techniques carries a high level of uncertainty and requires using a calibration standard with well-known thermal properties. In this work, we used 200-nm-thick titanium dioxide (TiO<sub>2</sub>) film<sup>5</sup> for calibration, with 49% transmission at 355 nm. These data provided an estimate (with 100% uncertainty) for the PHI sensitivity of 1 μV per 40 parts per million (ppm) absorbance, which is a ratio of absorbed power to incident power. HfO<sub>2</sub> film samples produced in different coating runs and used in this study showed some variation in initial

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355-nm absorption, with the majority of samples characterized by  $<400$ -ppm absorptance. A detailed description of the PHI technique principle and the setup used in this work is given in Refs. 2 and 5.

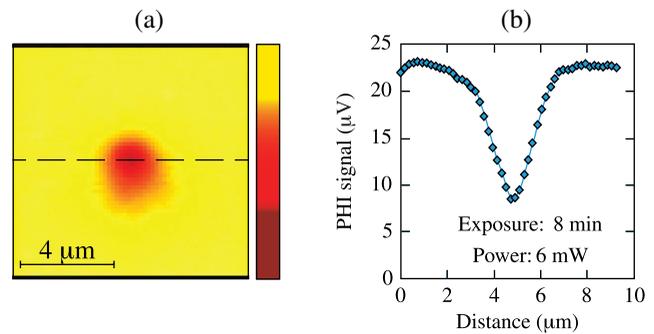
Near-UV cw laser-absorption annealing was studied using either a 351-nm, 1-W  $\text{Ar}^+$  laser or a 355-nm, diode-pumped semiconductor laser that works as a pump laser for PHI and delivers up to 6 mW on the sample. The latter, after being focused into a  $\leq 0.7\text{-}\mu\text{m}$  [full width at half maximum (FWHM)] spot by a  $40\times/0.95$  numerical-aperture objective, produced power densities of up to  $1\text{ MW}/\text{cm}^2$ , and the former was focused into a  $50\text{-}\mu\text{m}$  spot ( $1/e^2$ ), resulting in a  $46\text{-kW}/\text{cm}^2$  power density. A sample exposure, in the case of a small laser spot (PHI pump laser), was accomplished by using two different methods. In the first method, the sample position and laser power were fixed and the sample was exposed for some period of time, typically up to 15 min. In this manner, several sample sites were irradiated at a different cw laser power. The second method consisted of raster scanning the sample with typical velocities of 1 to  $10\text{ }\mu\text{m}/\text{s}$ , producing different exposures by varying the sample velocity and laser power. To quantify the exposure effect at a particular location, another raster scan of the larger area, including the exposed area, was performed with a high sample velocity and low laser power to minimize additional energy deposition. In the case of a large laser spot ( $\text{Ar}^+$  laser), the sample was translated in one direction with a  $1\text{-}\mu\text{m}/\text{s}$  velocity, creating a  $50\text{-}\mu\text{m}$ -wide, several-millimeters-long exposed area.

Laser-damage testing of cw laser exposed and unexposed sample areas was performed in a 1-on-1 regime, using 351-nm, 0.9-ns pulses of a Nd:YLF diode-pumped laser<sup>6</sup> and 1053-nm, 600-fs, best-compression pulses from a laser using the standard chirped pulse amplification scheme.<sup>7</sup> Damage testing using 0.9-ns pulses was conducted in an ambient environment and testing with 600-fs pulses was performed in a  $10^{-7}$ -Torr vacuum to avoid self-focusing in air. The laser beam spot size ( $1/e^2$ ) on the sample was 400 and  $270\text{ }\mu\text{m}$  for 0.9-ns and 600-fs beams, respectively. Laser-damage morphology was investigated by means of atomic force microscopy (AFM).

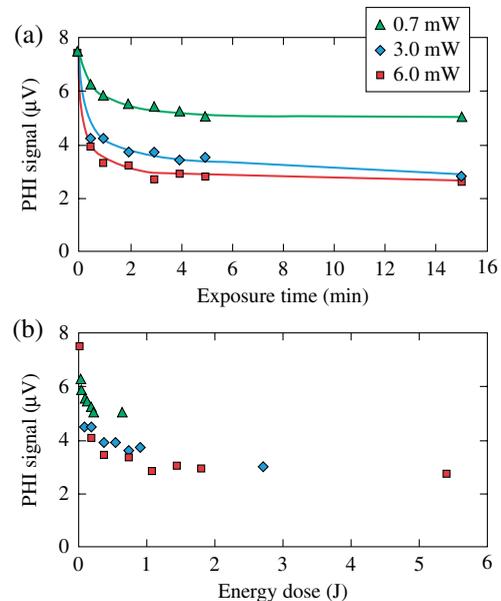
### 3 Results and Discussion

#### 3.1 Absorption-Annealing Effects

Figure 1 shows the result of irradiating the  $\text{HfO}_2$  film for 8 min at a fixed location using a 355-nm, 6-mW beam focused into a submicrometer spot ( $\leq 0.7\text{ }\mu\text{m}$ ,  $\sim 1\text{-MW}/\text{cm}^2$  power density). Figure 1(a) is a PHI scan of a  $10\times 10\text{-}\mu\text{m}^2$  film area centered around the location of the laser spot. A cross-sectional profile of the PHI signal [Fig. 1(b)] shows up to 70% reduction in absorption within the irradiated spot, which appeared to be permanent when confirmed by PHI scans performed after one week and then one month later. Next, absorption annealing was investigated as a function of laser power and exposure time. Figure 2(a) plots a normalized PHI signal's dependence on exposure time for three different values of laser-beam power: 0.7, 3, and 6 mW. One can see that the main drop in signal takes place during the first minute and is then followed by a slow decline on an  $\sim 10$ - to 15-min time scale. The initial signal drop becomes faster and deeper with increasing



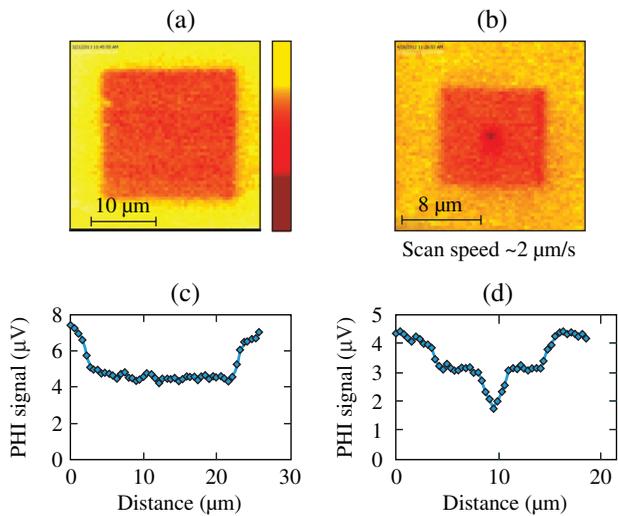
**Fig. 1** (a) A photothermal heterodyne imaging (PHI) map of a  $10\times 10\text{-}\mu\text{m}^2$  film area. (b) The signal horizontal profile through the central spot irradiated by a 355-nm, 6-mW cw laser for 8 min.



**Fig. 2** (a) Temporal behavior of absorption as a function of cw laser power on sample. (b) Normalized PHI signal for different laser powers as a function of energy dose (incident energy-time product).

laser power. Nevertheless, the temporal behavior of the 6- and 3-mW curves shows that at long exposures, the signal can eventually be stabilized at the same level. It suggests that within some range of power densities, overall absorption reduction is proportional to both power density and exposure time or, in different terms, locally deposited energy. This notion is strongly supported by the data presented in Fig. 2(b), showing the PHI signal plotted as a function of energy dose (incident power-time product). One can see that the data for different laser powers merge together, thereby confirming that the deposited energy (which is proportional to the energy dose) is a defining factor for producing absorption annealing effect.

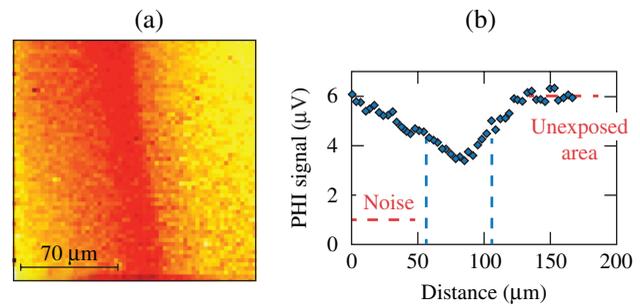
In an attempt to demonstrate the possibility of annealing absorption within a sample area larger than the laser spot, we performed square raster scans with linear dimensions of several tens of micrometers. Figure 3 shows corresponding PHI images, each obtained as a result of two scans. In the first image [Fig. 3(a)], the central  $20\times 20\text{-}\mu\text{m}$  part was initially scanned with a high laser power of 4.5 mW and a low scan velocity of  $\sim 1\text{ }\mu\text{m}/\text{s}$  to achieve the annealing



**Fig. 3** PHI maps resulting from 355-nm cw laser annealing by means of raster scanning with a low, 1- $\mu\text{m/s}$  sample velocity: (a) a  $30 \times 30\text{-}\mu\text{m}^2$  area with the central  $20 \times 20\text{-}\mu\text{m}^2$  part exposed using 4.5-mW laser power; (b) a  $20 \times 20\text{-}\mu\text{m}^2$  area with the central  $10 \times 10\text{-}\mu\text{m}^2$  part raster scan exposed, using 6 mW and, in addition, the central spot exposed for 8 min with 6-mW power; (c) and (d) horizontal signal profiles taken through the central part of (a) and (b) maps, respectively.

effect. Subsequently, the larger square area ( $30 \times 30 \mu\text{m}^2$ ), with the same center coordinate, was imaged using a much lower laser power of 1.5 mW and a several-times-higher speed velocity, such that no, or very little, annealing effect was produced by the second scan. A cross-section horizontal signal profile revealed at least a 40% reduction in absorption within the initially scanned area. A similar procedure was used for the  $20 \times 20\text{-}\mu\text{m}$  image shown in Fig. 3(b), with the only difference being that the central spot was exposed for an additional 8 min using a 6-mW laser power. In this case, the horizontal signal profile shows a 70% reduction in absorption in the central spot. From a practical point of view, it should be noted that because of the scan velocity and beam size limitations (the latter defines the maximum separation between two consecutive scan lines, or minimum number of lines per scan), raster scanning for annealing purposes is very time consuming. For example, it takes at least 2 h to complete a  $60 \times 60\text{-}\mu\text{m}$  scan using a 1- $\mu\text{m/s}$  scan velocity.

The question to be addressed is the possibility of scaling up the absorption-annealing process for  $\text{HfO}_2$  films used in optical parts for laser applications. In this work, we explored the possibility of producing absorption annealing in a  $\text{mm}^2$ -scale area using a cw  $\text{Ar}^+$  laser having a maximum 351-nm output power of 1 W. In this case, a laser beam having a power of  $\sim 900$  mW was focused into a  $50\text{-}\mu\text{m}$ -diameter ( $1/e^2$ ) spot on the sample, which was slowly ( $\sim 1 \mu\text{m/s}$ ) linearly translated for a distance of 3.6 mm. Despite a much lower power maximum density of  $46 \text{ kW/cm}^2$  as compared to  $\sim 1 \text{ MW/cm}^2$  in the case of a small PHI pump laser spot, a much longer (at least 50 $\times$ ) exposure time allowed us to achieve an  $\sim 50\%$  absorption reduction (see Fig. 4) in the film area of  $\sim 0.2 \text{ mm}^2$ . In this exposure regime, the energy dose per unit of film area is comparable with an  $\sim 0.5$ -min exposure using a small beam with a power of 0.7 mW (see Fig. 2).



**Fig. 4** (a) A 180- $\mu\text{m}$  PHI scan. (b) Corresponding horizontal signal profile of the hafnium oxide ( $\text{HfO}_2$ ) film area irradiated by a 900-mW  $\text{Ar}^+$  laser. The sample vertical travel velocity was  $1 \mu\text{m/s}$ .

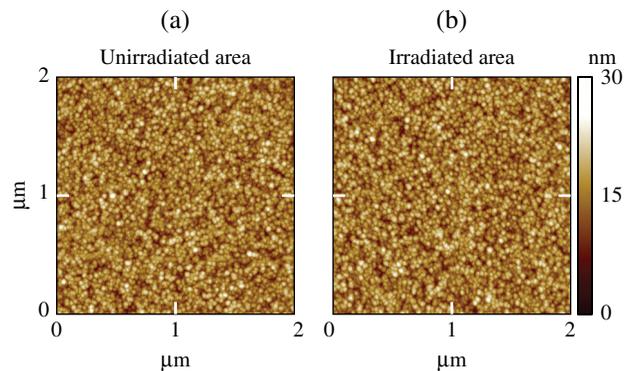
### 3.2 Laser-Damage Performance of Annealed $\text{HfO}_2$ Films

#### 3.2.1 Nanosecond-pulse damage

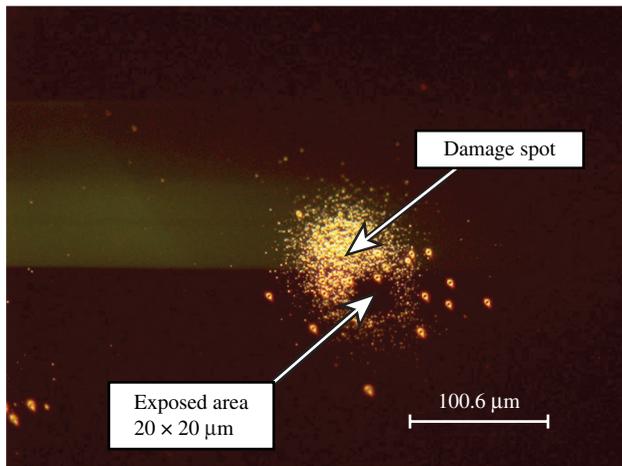
Laser-damage performance of thin films is usually strongly linked to the absorption properties of the film material and, therefore, can provide a true measure of absorption annealing. In this work, damage thresholds and damage morphology were investigated for cw laser-annealed film areas and then compared to the damage behavior of unexposed, as-produced film areas. As a starting point, we conducted AFM imaging of the cw laser-annealed film columnar structure, which was then compared to the columnar structure of the unexposed film. High-resolution ( $\sim 7\text{-nm}$ ) AFM images of these two areas (see Fig. 5) did not reveal any modification caused by the near-UV, cw laser exposure with power densities up to  $\sim 1 \text{ MW/cm}^2$ , implying that local heating of the material produced temperatures well below the  $\text{HfO}_2$  melting point.

To evaluate the effect of absorption annealing on  $\text{HfO}_2$  film-damage resistance, a series of  $20 \times 20\text{-}\mu\text{m}$  cw laser-exposed areas were produced on a sample and then irradiated by a pulsed laser at fluences exceeding the damage threshold. Figure 6 depicts an optical micrograph of such a film area irradiated by a 351-nm, 0.9-ns pulse with a peak fluence  $\sim 30\%$  above the threshold. A square-shaped unaffected area where cw-laser exposure was carried out is clearly visible inside the damaged zone. Laser-fluence estimates show an  $\sim 25\%$  increase in damage threshold within the film area subjected to annealing, which unambiguously proves that absorption is reduced in cw laser-exposed film.

A high-resolution AFM map (see Fig. 7) of the sample site, shown earlier in Fig. 6, provides additional information



**Fig. 5** Atomic force microscopy (AFM) images ( $2 \times 2 \mu\text{m}$ ) of cw laser, unirradiated (a) and irradiated (b)  $\text{HfO}_2$  film areas.

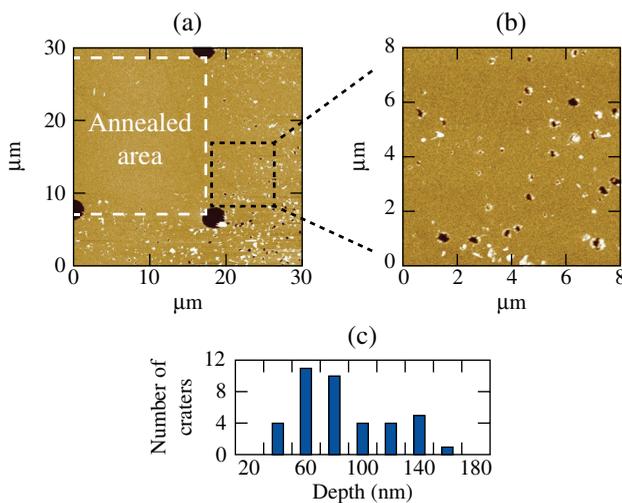


**Fig. 6** An optical micrograph of an HfO<sub>2</sub> film site damaged by a 351-nm, 0.9-ns single pulse. Damage morphology clearly shows a damage-free, ~20 × 20-μm<sup>2</sup> square area subjected to exposure by a 6-mW, 355-nm cw laser.

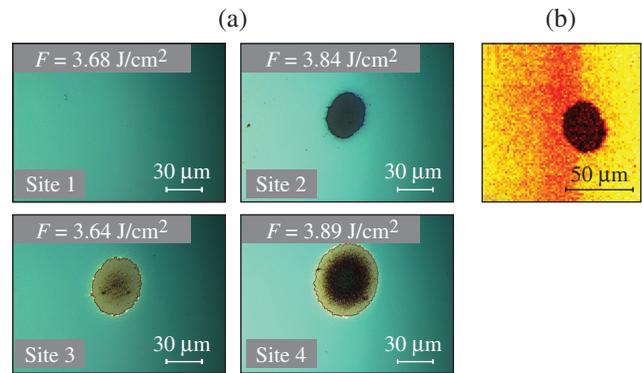
of the impact of annealing on absorption sources in film material. Taking into account that damage morphology is represented by isolated craters, crater depth distribution provides a rough approximation for the localized absorber distribution within the sample material. The crater depth distribution was measured for the damage site area adjacent to the exposed area [Fig. 7(b)], which should provide a reasonable estimate of the initial absorber distribution within the exposed material. One can see that the crater depth does not exceed the 180-nm depth that equals the thickness of the HfO<sub>2</sub> layer. This result indicates that annealed absorption precursors are indeed located inside the HfO<sub>2</sub> film and not in the SiO<sub>2</sub> film or the substrate.

### 3.2.2 Short-pulse (600-fs) damage

Previous studies<sup>8</sup> suggest that electronic defects might also play a role in short-pulse (picosecond, femtosecond) laser



**Fig. 7** (a) A 30 × 30-μm<sup>2</sup> AFM scan of a nanosecond-pulse damaged film site, including part of a cw laser-exposed area. (b) A higher-resolution, 8 × 8-μm scan showing damage craters bordering an undamaged cw laser-irradiated area. (c) Crater-depth distribution confirming the location of absorption sources inside the 180-nm-thick HfO<sub>2</sub> film.



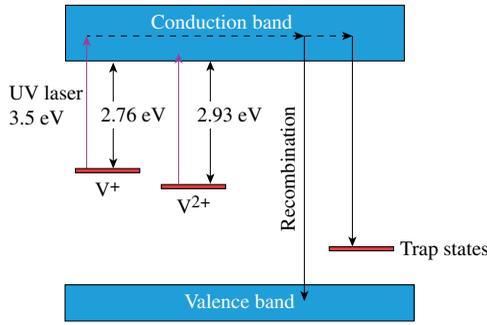
**Fig. 8** (a) Optical micrographs of 600-fs pulse-damage morphology. Sites 1 and 2: cw laser irradiated; sites 3 and 4: unirradiated. (b) PHI scan of site 2 showing relative positions of the cw laser-irradiated area and damaging-pulse imprint.

damage. In this context, the possible impact of the cw laser annealing of the absorption precursors on the short-pulse damage performance is of interest. Similar to the nanosecond-pulse study, near-UV, cw laser irradiation of a HfO<sub>2</sub> film was conducted utilizing both a small spot ( $\leq 0.7 \mu\text{m}$ ) of a PHI laser and a 50-μm spot of an Ar<sup>+</sup> laser. The results of the 1053-nm, 600-fs pulse damage testing of exposed (sites 1 and 2) and unexposed (sites 3 and 4) film areas are presented in Fig. 8, showing damage morphologies as recorded by an optical microscope. It is evident that the exposed sites show no damage at all while being irradiated by pulses with fluences above the unexposed film threshold of 3.45 J/cm<sup>2</sup> (compare sites 1 and 3) or show a smaller extent of damage (site 2 versus site 4) as compared to the unexposed sites. Moreover, as evidenced by the PHI image of damaged site 2 [see Fig. 8(b)], the damaging pulse has partially missed the exposed area, which, in the case of a better overlap, could show an even larger difference in the damage scale for sites 2 and 4.

To summarize, near-UV, cw laser annealing improves the damage resistance of *e*-beam-deposited HfO<sub>2</sub> films to pulsed laser radiation.

### 3.3 Absorption-Annealing Mechanisms

Mechanisms of the near-UV, cw laser annealing of absorption in HfO<sub>2</sub> films and related improvements in pulsed laser-damage resistance may be explained if one considers electronic defects as a main source of absorption and damage initiation. Numerous types of defects can exist in HfO<sub>2</sub> bulk material with electronic energy levels located inside the bandgap<sup>9</sup> and even more are expected to exist for HfO<sub>2</sub> films caused by the columnar film structure. As suggested earlier,<sup>2</sup> in the case of near-UV nanosecond-pulse damage, some of these states—as single- and double-ionized oxygen vacancies [V<sup>+</sup>, V<sup>2+</sup>, respectively (see Fig. 9)]—are shallow enough that absorption of 351-nm photons (3.54 eV) can initiate transition of the electron into the conduction band. Further heating of these free electrons by the remaining laser pulse energy can promote electron avalanche formation and damage. The same defect energy levels might initiate multiphoton absorption and damage in the case of short, 600-fs pulses at 1053 nm. Assuming the validity of such a damage mechanism, the cw laser-induced absorption-annealing effect and linked increase in pulsed laser-damage



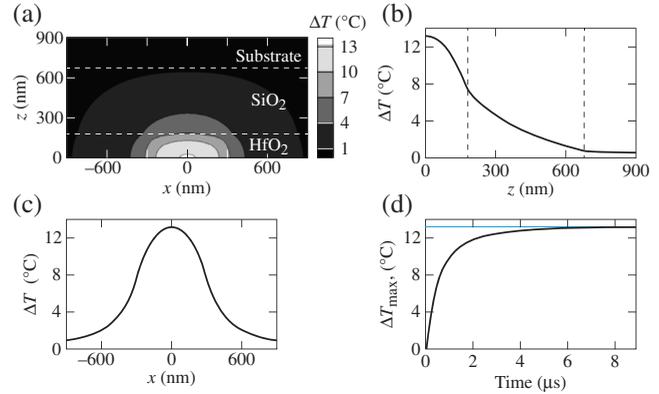
**Fig. 9** HfO<sub>2</sub> energy-level diagram illustrating the possible mechanism of near-UV absorption annealing.

resistance can be explained by depopulation of the absorbing states. The first possible scenario might involve cw laser-excited electron transition into the conduction band, where the electron spends time on the order of 10 ps (Ref. 8), followed by recombination with holes in the valence band or trapping into the deep defect states in the bandgap, as shown in Fig. 9. Modeling this scenario using kinetic equations may provide further clarification of the annealing mechanism. This type of study, as well as extending the investigation of cw laser annealing from monolayers to multilayer systems, should become the subject of future research.

The second absorption-annealing scenario may be linked to the heating of the film material resulting from the absorption of UV-laser photons. Thermal annealing is widely used to improve the mechanical and optical performance of thin films, including laser-damage resistance in the near-UV, as was recently demonstrated<sup>10</sup> for HfO<sub>2</sub> monolayer films at a 355-nm wavelength. In that work, the HfO<sub>2</sub> film temperature was increased by at least 100°C above room temperature in order to observe the absorption-annealing effect, with the maximum effect obtained at an annealing temperature of 300°C. To evaluate possible thermal effects in our study, the temperature distribution in the HfO<sub>2</sub> film was modeled with the assumption that all of the energy absorbed in the film is released in the form of heat. The energy deposition was considered homogeneous in the cylindrical film volume with a diameter equal to the FWHM diameter of the laser beam (~600 nm) and a height equal to a 180-nm film thickness. The geometry of the model is shown in Fig. 10(a). The cw laser intensity was fixed at 1 MW/cm<sup>2</sup> (highest used in the experiment), and the energy deposited in the film was varied through variation of the film absorption in the range of 10 to 1000 ppm, with the upper absorption boundary (1000 ppm) being well above the film absorption estimated from photothermal measurements. Heat conduction was considered to be the only channel of energy dissipation, and temperature rise in the film was obtained by solving appropriate heat-conduction equations.

$$\rho C \frac{\partial T}{\partial t} = \nabla \kappa \nabla T + S, \quad (1)$$

where  $\rho$ ,  $C$ , and  $\kappa$  are the density, heat capacity, and heat conductivity of the materials, respectively;  $T$  is the temperature; and  $S$  is the absorbed laser power per unit volume. The material parameters for modeling are listed in Table 1. No reliable data are available for the heat capacity and density



**Fig. 10** Peak temperature rise distribution in a thin-film sample: (a) modeling geometry and distribution in  $x$ - $z$  plane ( $y = 0$ ); (b) lineout along  $z$  axis ( $y = x = 0$ ); (c) lineout along  $x$  axis ( $y = z = 0$ ); and (d) time evolution of the maximum temperature rise.

values for HfO<sub>2</sub> and SiO<sub>2</sub> films; for that reason, their values were set equal to the bulk values. It should be noted here that these parameters in the cw laser regime affect only the time necessary to reach peak temperature but not the peak temperature value [as can be seen from Eq. (1) by setting the temperature derivative to zero]. A standard two-dimensional cylindrical (five-point spatial), time-explicit finite-difference scheme was used to discretize and solve the heat-conduction equation (see, e.g., chapter 8 of Ref. 11). The zero heat-flux boundary condition was applied at the film/air interface ( $z = 0$ ), and a zero temperature-change boundary condition was applied at the other boundaries of the computational domain, which was chosen to be large enough to not affect the calculated temperature by more than 0.1°C. The results of the modeling presented in Fig. 10 show that at the highest chosen absorption level of 1000 ppm, the peak temperature rise does not exceed 14°C, which is not enough to produce the absorption-annealing effect. The two-dimensional temperature change distribution is shown in Fig. 10(a) along with its lineouts along the axis of symmetry  $z$  ( $x = y = 0$ ) in Fig. 10(b) and along the  $x$  axis ( $y = z = 0$ ) in Fig. 10(c). The time evolution of the maximum temperature rise, which is observed along the center of the laser beam at the HfO<sub>2</sub>/air interface,  $x = y = z = 0$ , after the laser is turned on at time  $t = 0$ , is shown in Fig. 10(d). It is worth noting that all these conditions lead to the highest-possible peak film temperature estimates. Adding energy dissipation channels (such as luminescence or structural transformations) would lead to lower peak temperatures. Consequently, one can conclude that depopulation of the absorbing states is not caused by a heat-induced structural matrix transformation.

**Table 1** Parameters used to calculate the rise in hafnium oxide (HfO<sub>2</sub>) film temperature.

	Thermal conductivity (W/m · K)	Heat capacity (J/kg · K)	Density (kg/m <sup>3</sup> )
HfO <sub>2</sub> film	0.10	287	9700
Silicon dioxide film	0.25	741	2200
Fused-silica substrate	1.38	741	2200

Finally, it should be noted that irradiation of thin films by cw lasers with power densities used in this work can produce different absorption-modification effects. For instance, in similar experimental conditions,<sup>12</sup> irradiation of TiO<sub>2</sub> monolayer films using an 800-nm cw laser caused an increase in absorption that was also exposure-time dependent. Consequently, absorption-modification effects are thin-film material and cw laser wavelength specific.

#### 4 Conclusions

Irradiation of e-beam-deposited HfO<sub>2</sub> monolayer films by near-UV, cw laser light with power densities of 50 kW/cm<sup>2</sup> to 1 MW/cm<sup>2</sup> produced significant modification of near-UV film absorption. As high as a 70% reduction in absorption was achieved in film areas subjected to exposure. The effect is permanent, as confirmed by repeated measurements over a several-month period. It is shown that absorption modification is achieved without any changes in the film columnar structure on a spatial scale of up to a few nanometers. This led to the conclusion that reduction of absorption is linked to the modification of the atomic film structure—in particular, to changes in the concentration of structural defects responsible for near-UV absorption. Investigation of 351-nm, nanosecond-pulse laser-damage behavior of HfO<sub>2</sub> monolayer films subjected to cw laser exposure showed a >25% increase in damage thresholds and confirmed the reduction of absorption through annealing of absorbing defects residing inside the HfO<sub>2</sub> film. A 600-fs pulse irradiation also indicated an increase in laser-damage resistance of cw laser-exposed areas as compared to unexposed areas. This result suggests partial cw laser annealing of electronic defects participating in the multiphoton absorption process that initiates damage by 600-fs pulses. Future research will concentrate on further clarification of the annealing mechanism and will extend the study to multilayer systems.

#### Acknowledgments

This material is based upon work supported by the Department of Energy National Nuclear Security Administration under Award Number DE-NA0001944, the University of Rochester, and the New York State Energy Research and Development Authority. The support of DOE does not constitute an endorsement by DOE of the views expressed in this article.

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